

# Mg–Zr Cosubstituted Ta<sub>3</sub>N<sub>5</sub> Photoanode for Lower-Onset-Potential Solar-Driven Photoelectrochemical Water Splitting

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## S Supporting Information

**ABSTRACT:** In p/n photoelectrochemical (PEC) cell systems, a low onset potential for the photoanode, as well as a high photocurrent, are critical for efficient water splitting. Here, we report a Mg–Zr cosubstituted Ta<sub>3</sub>N<sub>5</sub> (Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr) photoanode, designed to provide a more negative onset potential for PEC water splitting. The anodic photocurrent onset on Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr was 0.55 V<sub>RHE</sub> under AM 1.5G-simulated sunlight, which represented a negative shift from the ca. 0.8 V<sub>RHE</sub> for pure Ta<sub>3</sub>N<sub>5</sub>. This negative shift in the onset potential of PEC water splitting was attributed to the change in the bandgap potential due to partial substitution by the foreign ions Mg<sup>2+</sup> and/or Zr<sup>4+</sup>.

Water splitting into hydrogen and oxygen by solar energy is an attractive way to produce clean and renewable energy. Photoelectrochemical (PEC) water splitting on semiconductor electrodes is expected to become a widespread means of efficient solar energy conversion. For PEC water splitting, several cell configurations have been proposed:<sup>1</sup> half-cells (p-type/n-type photoelectrode and a counter electrode) assisted by an external bias and p/n PEC cells electrically combining photocathodes with photoanodes. Among these, p/n PEC cells are considered to be a more efficient route for unassisted water splitting. Despite the remarkable recent progress in the development of p-type photocathodes,<sup>2,3</sup> many challenging tasks still remain in n-type photoanodes. Specifically, the onset potential of the photoanodic current in PEC water splitting is too positive, which hampers the realization of p/n PEC water splitting devices. Therefore, the development of efficient photoanodes with a more negative photocurrent onset is highly desirable.

Several nitrides and oxynitrides (e.g., Ta<sub>3</sub>N<sub>5</sub>, LaTaON<sub>2</sub>, LaTiO<sub>2</sub>N, BaTaO<sub>2</sub>N, etc.) with optical absorption at 600 nm or longer have been identified as attractive semiconductor photocatalysts/photoanodes in terms of having suitable bandgaps for splitting water under visible light.<sup>4–7</sup> In particular, Ta<sub>3</sub>N<sub>5</sub>, which has an anosovite (Ti<sub>3</sub>O<sub>5</sub>) structure consisting of corner- and edge-sharing TaN<sub>6</sub> octahedra, is a promising candidate for efficient PEC water splitting. Owing to its simple composition, there are various routes to fabricating a photoanode

using this material. Thus, the PEC activity of Ta<sub>3</sub>N<sub>5</sub> photoanodes has been improved by incorporating different structures such as particulates, nanorods, and thin films, and by modifying the surface of the electrode with a cocatalyst.<sup>4,8–10</sup> The photoanodic current at 1.23 V<sub>RHE</sub> increased significantly by those attempts, whereas the photocurrent onset remained more positive than 0.8 V<sub>RHE</sub> in most cases. The high onset potential, indicative of poor photoactivity, suggests the limitation of pure Ta<sub>3</sub>N<sub>5</sub> as a photoanode.

Compositional modification by introducing foreign ions is a typical strategy to adjust semiconductor properties, which has led to enhanced photoelectrolysis/catalysis in many successful cases. For example, doping of the Ta<sub>3</sub>N<sub>5</sub> electrode with Na<sup>+</sup> or Ba<sup>2+</sup> was reported to enhance PEC water splitting.<sup>11,12</sup> Since 6-coordinate Na<sup>+</sup> (102 pm) and Ba<sup>2+</sup> (135 pm) have significantly larger ionic radii than 6-coordinate Ta<sup>5+</sup> (64 pm), the isostructural replacement of Ta<sup>5+</sup> with Na<sup>+</sup> or Ba<sup>2+</sup> in the anosovite structure is crystallographically unfavorable and leads to phase separation under heavy doping.<sup>11,13</sup> Mg<sup>2+</sup> (72 pm) and Zr<sup>4+</sup> (72 pm), with ionic radii comparable to that of Ta<sup>5+</sup>, are suitable foreign ions for forming Ta<sub>3</sub>N<sub>5</sub> variants. Owing to the difference in valence between the foreign and parent cations, the substitution of Mg<sup>2+</sup> or Zr<sup>4+</sup> for Ta<sup>5+</sup> concurrently causes N<sup>3–</sup>/O<sup>2–</sup> replacement to maintain the charge balance, creating an oxynitride, Ta<sub>3–x</sub>Mg<sub>x</sub>N<sub>5–3x</sub>O<sub>3x</sub> or Ta<sub>3–x</sub>Zr<sub>x</sub>N<sub>5–x</sub>O<sub>x</sub>, respectively. Thus, isostructural substitution is feasible for these foreign ions over a wide compositional range.<sup>14,15</sup>

In the present work, we fabricated Mg- and/or Zr-substituted Ta<sub>3</sub>N<sub>5</sub> electrodes by the particle transfer (PT) method<sup>5</sup> with the appropriate powder samples, exploiting the flexibility in the fabrication route. The prepared Ta<sub>3</sub>N<sub>5</sub>-variant photoanodes were evaluated for water splitting photoactivity through various PEC measurements. Here, we report that the cointroduction of foreign Mg<sup>2+</sup> and Zr<sup>4+</sup> into Ta<sub>3</sub>N<sub>5</sub> at high substitution levels caused an obvious negative shift in the onset potential for PEC water oxidation. In addition, the effect of Mg and/or Zr substitution on PEC activity will be explained on the basis of band gap potentials. Photocatalysis of the Ta<sub>3</sub>N<sub>5</sub> variants in

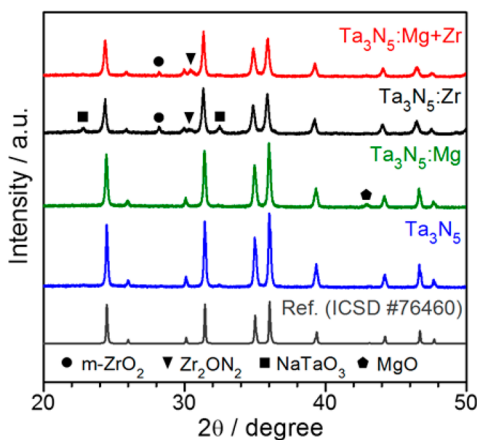
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powder form will also be discussed to support their enhanced PEC activities relative to that of pure  $\text{Ta}_3\text{N}_5$ .

The powder  $\text{Ta}_3\text{N}_5$  variants were synthesized by flux-assisted nitridation using the corresponding oxide precursors along with a  $\text{Na}_2\text{CO}_3$  flux under  $\text{NH}_3$  flow at 1173 K. The oxide precursors were prepared by heating  $\text{Ta}_2\text{O}_5$  mixed with  $\text{ZrO}(\text{NO}_3)_2 \cdot 2\text{H}_2\text{O}$  and/or  $\text{Mg}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  at 923 K in air. The ratio of Mg:Zr in the cosubstituted photocatalysts is 1:2 unless otherwise noted. The synthesized  $\text{Ta}_3\text{N}_5$  photocatalysts replaced with no ions,  $\text{Mg}^{2+}$ ,  $\text{Zr}^{4+}$ , and  $\text{Mg}^{2+} + \text{Zr}^{4+}$  will be denoted as  $\text{Ta}_3\text{N}_5$ ,  $\text{Ta}_3\text{N}_5:\text{Mg}$ ,  $\text{Ta}_3\text{N}_5:\text{Zr}$ , and  $\text{Ta}_3\text{N}_5:\text{Mg}+\text{Zr}$ , respectively. For comparison, a typical  $\text{Ta}_3\text{N}_5$  sample (hereafter “ $\text{Ta}_3\text{N}_5$  (no flux)”) was also nitrided from  $\text{Ta}_2\text{O}_5$  in the absence of a  $\text{Na}_2\text{CO}_3$  flux.<sup>16</sup> The photoelectrodes, which were fabricated by the PT method, consisted of an absorber particle layer, Nb ohmic contact layer, and Ti current collector layer. Subsequently, a bilayered Co–Fe oxide (hereafter “ $\text{CoO}_x\text{-FeO}_x$ ”) as cocatalyst was deposited on the photoanodes by pulse-current electrodeposition. PEC water splitting was performed in a conventional three-electrode system under simulated sunlight irradiation (AM1.5G). Further details of the experimental procedures are given in the Supporting Information (SI).

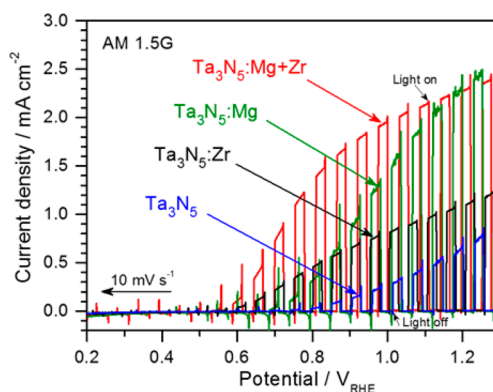
The crystal structures of the synthesized  $\text{Ta}_3\text{N}_5$ ,  $\text{Ta}_3\text{N}_5:\text{Mg}$ ,  $\text{Ta}_3\text{N}_5:\text{Zr}$ , and  $\text{Ta}_3\text{N}_5:\text{Mg}+\text{Zr}$  were determined by XRD in Figure 1. The replacement of  $\text{Ta}^{5+}$  with  $\text{Mg}^{2+}$  and/or  $\text{Zr}^{4+}$  was fixed to 33



**Figure 1.** XRD patterns for  $\text{Ta}_3\text{N}_5$ ,  $\text{Ta}_3\text{N}_5:\text{Mg}$ ,  $\text{Ta}_3\text{N}_5:\text{Zr}$ , and  $\text{Ta}_3\text{N}_5:\text{Mg}+\text{Zr}$  photocatalysts synthesized by nitriding at 1173 K for 20 h under  $\text{NH}_3$  flow.

at %, which was optimized by PEC water splitting activity. All photocatalysts showed XRD patterns assignable to the anosovite ( $\text{Ti}_3\text{O}_5$ ) structure, but the  $\text{Ta}_3\text{N}_5$  variants that underwent elemental substitution also exhibited small amounts of segregated phases such as  $\text{ZrO}_2$ ,  $\text{Zr}_2\text{ON}_2$ ,  $\text{NaTaO}_3$ , and  $\text{MgO}$ . The (110) peaks of  $\text{Mg}^{2+}$  and/or  $\text{Zr}^{4+}$ -substituted  $\text{Ta}_3\text{N}_5$  slightly shifted toward lower angles as compared to that of  $\text{Ta}_3\text{N}_5$ , indicating the small lattice expansion by increased average ionic radii (Figure S1 in SI). ICP and O/N analysis revealed that the cosubstitution of Mg and Zr was accompanied by the O/N substitution to compensate an imbalance of ionic charge (Table S1 in SI). Although  $\text{Mg}^{2+}$  and  $\text{Zr}^{4+}$  have 12% larger ionic radii than  $\text{Ta}^{5+}$ , the concurrent substitution of smaller  $\text{O}^{2-}$  for  $\text{N}^{3-}$  reduced the extent of lattice expansion.

Figure 2 and Figure S2 in SI show the photocurrent–potential ( $I$ – $E$ ) curves for various  $\text{Ta}_3\text{N}_5$  variant electrodes modified with and without  $\text{CoO}_x\text{-FeO}_x$ , respectively, under chopped AM1.5G



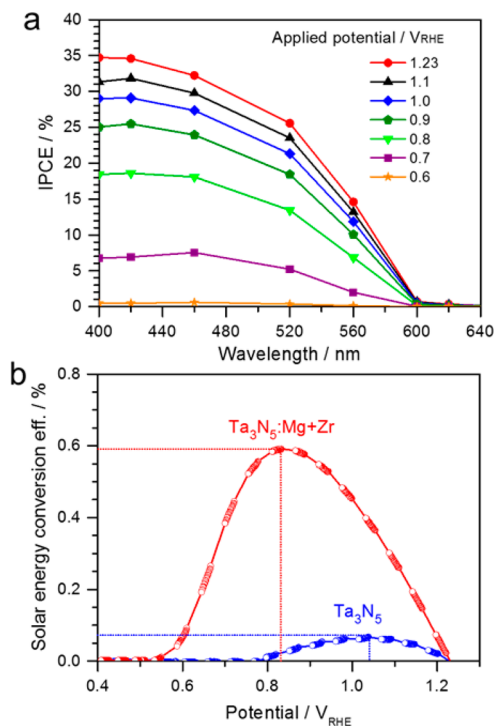
**Figure 2.** LSVs for  $\text{CoO}_x\text{-FeO}_x/\text{Ta}_3\text{N}_5$ ,  $\text{Ta}_3\text{N}_5:\text{Mg}$ ,  $\text{Ta}_3\text{N}_5:\text{Zr}$ , and  $\text{Ta}_3\text{N}_5:\text{Mg}+\text{Zr}$  photoanodes toward PEC water splitting under chopped AM1.5G sunlight. The measurements were performed by cathodically sweeping the potential from 1.3 to 0.2  $\text{V}_{\text{RHE}}$  at a scan rate of  $10 \text{ mV s}^{-1}$  in a stirred Ar-saturated 0.1 M aqueous  $\text{Na}_2\text{SO}_4$  electrolyte adjusted to pH 13.

sunlight. The anodic photocurrent was generated on the  $\text{Ta}_3\text{N}_5$  electrode at around  $0.82 \text{ V}_{\text{RHE}}$  and gently increased above this potential, reaching  $0.72 \text{ mA cm}^{-2}$  at  $1.23 \text{ V}_{\text{RHE}}$ .  $\text{Ta}_3\text{N}_5$  (no flux) exhibited an almost identical photocurrent onset at  $0.78 \text{ V}_{\text{RHE}}$  but a higher photocurrent (Figure S3 in SI). The lower photocurrent for the  $\text{Ta}_3\text{N}_5$  would be attributable not only to the contamination of flux after the nitridation but also to different morphology of the synthesized photocatalysts (Figure S4 in SI). Thus, irrespective of the photocurrent, the onset potential was determined to be ca.  $0.80 \text{ V}_{\text{RHE}}$ , which likely reflected the inherent PEC activity of  $\text{Ta}_3\text{N}_5$  prepared by the PT method. On the  $\text{Ta}_3\text{N}_5:\text{Zr}$  electrode, the onset potential was observed at  $0.57 \text{ V}_{\text{RHE}}$ , which clearly represented a shift toward more negative potential relative to that for the  $\text{Ta}_3\text{N}_5$  electrode. The  $\text{Ta}_3\text{N}_5:\text{Mg}$  electrode generated an anodic photocurrent above  $0.70 \text{ V}_{\text{RHE}}$ , which represents a relatively small negative shift in onset potential. However, at  $1.23 \text{ V}_{\text{RHE}}$  it generated a photocurrent twice as high as that for  $\text{Ta}_3\text{N}_5:\text{Zr}$ . The  $\text{Ta}_3\text{N}_5:\text{Mg}+\text{Zr}$  photoanode exhibited a lower onset potential at  $0.55 \text{ V}_{\text{RHE}}$  and a photocurrent of  $2.3 \text{ mA cm}^{-2}$  at  $1.23 \text{ V}_{\text{RHE}}$ . Note that the onset for  $\text{Ta}_3\text{N}_5:\text{Mg}+\text{Zr}$  was almost identical to that for the  $\text{Ta}_3\text{N}_5:\text{Zr}$  electrode, and its photocurrent increased more steeply than those for  $\text{Ta}_3\text{N}_5:\text{Mg}$  and  $\text{Ta}_3\text{N}_5:\text{Zr}$ . The cosubstituted  $\text{Ta}_3\text{N}_5$  nitrided with no flux was prepared for comparison. The lower PEC activity revealed that the flux-assisted nitridation was very effective to synthesize the heavily substituted  $\text{Ta}_3\text{N}_5$  with higher crystallinity (Figure S5 in SI). Moreover, a different Mg:Zr ratio (= 2:1) was examined, and it was found that the PEC activity was almost unchanged (Figure S6 in SI). The significantly different PEC activity induced by Mg and Zr cosubstitution points to a synergetic effect of the two single substitutions: Zr causes a negative shift in the onset potential, while Mg increases the photocurrent.

The cosubstitution of  $\text{Mg}^{2+}$  and  $\text{Zr}^{4+}$  enhanced the PEC activity over the compositional range of 14–43 at %, demonstrating the effectiveness of heavy substitution (Figure S7 in SI). Moreover,  $\text{Ta}_3\text{N}_5:\text{Mg}+\text{Zr}$  with 33 at % substitution showed the highest PEC activity among the various compositions examined. Clearly, the observed enhancements were not attributable to the byproduct phases originating from Mg and Zr additives (Figure S8 in SI). The UV–vis DRS spectra for the  $\text{Ta}_3\text{N}_5$  variants showed that the optical absorption edge of each

powder was almost unchanged regardless of the type of substitution (Figure S9 in SI). This is most likely because the constituent orbitals for the conduction and valence band edges are invariably Ta 5d and N 2p states, respectively.<sup>17</sup> This also ruled out the possibility that the enhanced PEC activities were caused solely by the changed bandgap energies. Microgas chromatography demonstrated that H<sub>2</sub> and O<sub>2</sub> gases evolved during PEC water splitting in the stoichiometric ratio of 2:1 and almost achieved a Faradaic efficiency of unity (Figure S10 in SI). This indicates that the anodic photocurrent indeed originated from the oxygen evolution reaction by water splitting.

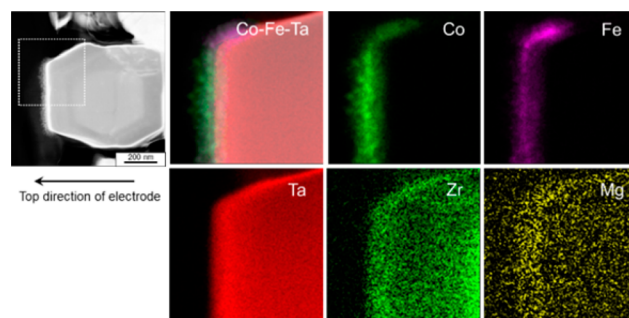
Figure 3a plots the wavelength dependence of the incident photon-to-current conversion efficiency (IPCE) on the



**Figure 3.** (a) Wavelength dependence of IPCE for the CoO<sub>x</sub>-FeO<sub>x</sub>/Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr photoanode measured at different applied potentials between 0.6 and 1.23 V<sub>RHE</sub>. (b) HC-STH conversion efficiencies (%) for the CoO<sub>x</sub>-FeO<sub>x</sub>/Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr and CoO<sub>x</sub>-FeO<sub>x</sub>/Ta<sub>3</sub>N<sub>5</sub> photoanodes.

Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr photoanode measured at different applied potentials. The onset wavelengths for the photocurrents agreed well with the light absorption edge of Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr. The IPCEs in the whole wavelength area increased significantly as the applied potential was increased from 0.6 to 0.8 V<sub>RHE</sub>, and increased moderately above this potential. The IPCEs at 0.8 V<sub>RHE</sub> were ca. 9–18% at 560–400 nm. In Figure 3b, half-cell solar-to-hydrogen (HC-STH) conversion efficiencies for the Ta<sub>3</sub>N<sub>5</sub> and Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr photoanodes are plotted as a function of applied potential, using the voltammograms shown in Figure 2. The HC-STH for Ta<sub>3</sub>N<sub>5</sub> (no flux) was also presented (Figure S11 in SI). The maximum HC-STH conversion efficiency for Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr was 0.59% at 0.82 V<sub>RHE</sub>, indicating that the Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr photoanode operated most effectively in the case of PEC water splitting around 0.8 V<sub>RHE</sub>. Upon substitution with Mg<sup>2+</sup> and Zr<sup>4+</sup>, the potential at which HC-STH was maximized shifted by 0.2 V<sub>RHE</sub> in the negative direction. This is favorable for unassisted water splitting using p/n PEC cells.

The microstructure of Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr was characterized by SEM analysis (Figure S4 in SI). Broadly speaking, two types of particle morphologies were observed: particles a few hundred nanometers in size and larger columnar particles. The columnar crystallites were observed only in the case of Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr, while the other samples consisted mostly of particles a few hundred nanometers in size. Moreover, HRTEM and SAED revealed clear lattice fringes and the corresponding Ta<sub>3</sub>N<sub>5</sub> diffraction pattern for a columnar particle, indicating the single crystallinity of the anosovite structure (Figure S12 in SI). Figure 4 shows a typical cross-sectional TEM image and the



**Figure 4.** Cross-sectional STEM image and elemental EDS mapping results for a single Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr particle loaded on a photoanode fabricated by the PT method. EDS mapping was conducted inside the white square in the STEM image. The elements Co and Fe derive from the cocatalyst CoO<sub>x</sub>-FeO<sub>x</sub> electrodeposited on the photoanode. Ta, Zr, and Mg represent the constituent elements of the Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr.

corresponding EDS mapping for the CoO<sub>x</sub>-FeO<sub>x</sub>/Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr electrode. The EDS elemental mapping demonstrated that Mg and Zr were homogeneously distributed inside a single particle, indicating that the foreign ions were incorporated into the Ta<sub>3</sub>N<sub>5</sub> crystal structure. It was also found that the bilayered CoO<sub>x</sub>/FeO<sub>x</sub> was deposited only on top of the particle, i.e., on the exposed surface of the photoelectrode. No large columnar particles were observed in the Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr nitride sample using three times the normal amount of Na<sub>2</sub>CO<sub>3</sub> flux, but the onset potential was identical (Figure S13 in SI). This suggests that the negative shift in the onset potential for PEC water splitting was mainly due to changes in the semiconducting properties of Ta<sub>3</sub>N<sub>5</sub> induced by the substitution, rather than to morphological changes.

The band positions and flat band potentials ( $E_{fb}$ ) of Ta<sub>3</sub>N<sub>5</sub> and Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr were investigated by Tauc plots, PESA spectra, and MS plots to discuss their semiconductor properties (Figure S14 in SI). A detailed discussion is presented in the Supporting Information. Figure 5 shows the expected band diagrams for Ta<sub>3</sub>N<sub>5</sub> and Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr. The conduction and valence band edges for Ta<sub>3</sub>N<sub>5</sub> were estimated to be -0.6 and 1.5 V<sub>NHE</sub> (pH = 0). However, Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr had conduction and valence band edges at -0.8 and 1.35 V<sub>NHE</sub> (pH = 0), respectively. The  $E_{fb}$  values for Ta<sub>3</sub>N<sub>5</sub> and Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr, which were determined to be -0.18 and -0.43 V<sub>NHE</sub>, respectively, shifted nearly in parallel to more negative onset potentials upon Mg-Zr cosubstitution. The negative shift in band gap potential is presumably due to the incorporation of more cationic ions (Mg<sup>2+</sup> and Zr<sup>4+</sup>) than Ta<sup>5+</sup>. It is thought that the observed negative shift in the energy levels contributed to the lower onset potential for the Ta<sub>3</sub>N<sub>5</sub>:Mg+Zr photoanode.

In addition, the photocatalytic activity of Ta<sub>3</sub>N<sub>5</sub> variants in powder suspension form was measured by conducting H<sub>2</sub> and O<sub>2</sub>



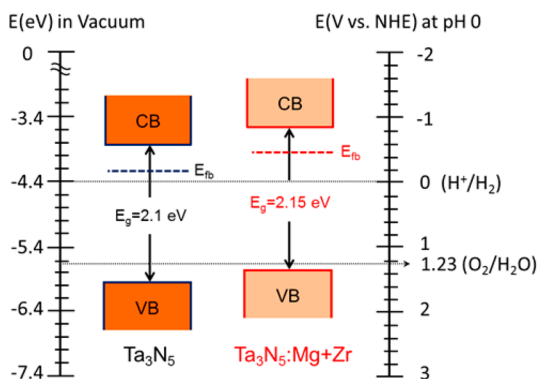


Figure 5. Band structure diagram of  $\text{Ta}_3\text{N}_5$  and  $\text{Ta}_3\text{N}_5\text{:Mg+Zr}$ .

evolution separately in the presence of MeOH and  $\text{Ag}^+$  as a sacrificial electron donor and acceptor, respectively. As seen in Figures S15 and S16 and Table S2 in SI, the photocatalytic activity changed markedly with the type of substitution. Mg and Zr single substitutions enhanced the  $\text{H}_2$  evolution several times, but reduced the  $\text{O}_2$  evolution capacity noticeably. The activity of  $\text{Ta}_3\text{N}_5\text{:Mg+Zr}$  (25 at%) for  $\text{H}_2$  evolution was approximately 15 times higher than that of  $\text{Ta}_3\text{N}_5$ , while the  $\text{O}_2$  evolution for the former was lower. These results corroborated the synergetic effect of the incorporated Mg and Zr, as discussed in the PEC examination above. Some of the elementary steps of photocatalysis are not identical to those of photoelectrolysis. However, the results of photocatalysis also indicate changes in semiconductor properties by compositional modification. The changes in  $\text{H}_2$  and  $\text{O}_2$  evolution capacity could be due to the observed shift in band position.

In summary, compositional modification of  $\text{Ta}_3\text{N}_5$  by partial substitution with foreign ions was demonstrated to be effective in enhancing PEC water splitting activity. Mg–Zr cosubstitution influenced the onset potential and photocurrent by a sort of synergetic effect so that a relatively large photocurrent at a more negative potential was obtained, recording a 0.59% HC-STH conversion efficiency at 0.82  $V_{\text{RHE}}$ . Although a dramatic improvement in photocurrent onset was achieved by compositional modification, the HC-STH value was still lower than the state-of-the-art Ba-doped  $\text{Ta}_3\text{N}_5$  nanorod photoanode (ca. 1.4% at the same potential).<sup>13</sup> The present flux-assisted nitridation method might be suitable for synthesizing a multicomponent oxynitride. Improving the synthesis method for multicomponent (oxy)nitride materials is key to further enhancing the photocurrent in addition to lowering the onset potential. The strategy of compositional modification demonstrated in the present study is expected to be applicable to a wide range of semiconductors and would enable the fabrication of more efficient photoelectrodes.

## ■ ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/jacs.5b08329.

Preparations of photocatalysts and photoanodes; physical characterization of photocatalysts; supplemental PEC activity; discussion of band structure (PDF)

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## Notes

The authors declare no competing financial interest.

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